

Application of Powder Materials and Functional Coatings
Применение порошковых материалов и функциональных покрытий

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Review article

Обзорная статья



Functional coatings of submersible oilfield equipment for protection against corrosion, asphalt, resin, paraffin and salt deposits: Review

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Abstract. The oil production process is often accompanied by various failures at oil production facilities, which leads to serious economic losses. Failures in oil production systems not only increase repair and maintenance costs, but also lead to loss of productivity, which has a negative impact on the economic efficiency of projects. A pipeline failure is considered to be its complete or partial shutdown due to a violation of its tightness or tightness of the shut-off valves, or due to blockage of the flow section. The most common causes of complications in oil production are: corrosion of oil and gas equipment, formation of asphalt-resin-paraffin deposits (ARPD) and inorganic salt deposits on the working surface of oil and gas equipment. There are a large number of methods aimed at preventing each of the previously mentioned complicating factors. It is noteworthy that the use of protective coatings can be a measure of prevention of corrosion processes, ARPD, and inorganic salt deposits. This article will review the literature, which will consider what properties, composition and structure protective coatings should have to prevent corrosion, ARPD and salt deposits, as well as what testing methods can be used to evaluate the ability of a protective coating to prevent these complicating factors.

Keywords: protective coatings, types of coatings, corrosion, asphalt-resin-paraffin deposits, salt deposits, coating testing methods

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Функциональные покрытия погружного
нефтепромыслового оборудования для защиты
от коррозии, асфальтосмолопарафиновых
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Аннотация. Процесс извлечения нефти часто сопровождается различными отказами на объектах нефтедобычи, что ведет к серьезным экономическим потерям. Отказы в системах нефтедобычи не только увеличивают затраты на ремонт и обслуживание, но также приводят к потере производительности, что негативно сказывается на экономической эффективности

проектов. Отказом трубопровода считается полная или частичная его остановка вследствие нарушения его герметичности или герметичности запорной арматуры, либо по причине закупорки проходного сечения. Наиболее распространенными причинами осложнений в нефтедобыче являются коррозия нефтегазового оборудования и образование асфальтосмолопарафиновых отложений (АСПО) и неорганических солейотложений на рабочей поверхности нефтегазового оборудования. Среди большого количества методов, направленных на предотвращение указанных осложняющих факторов, весьма эффективно применение защитных покрытий, которое может являться мерой профилактики и коррозионных процессов, и АСПО, и неорганических солейотложений. В настоящей статье проведен обзор литературных источников: рассмотрено, какими методами испытаний можно оценить способность защитного покрытия предотвращать возможные осложняющие факторы.

Ключевые слова: защитные покрытия, виды покрытий, коррозия, асфальтосмолопарафиновые отложения, солейотложения, методы испытания покрытий

Для цитирования: Юдин П.Е. Функциональные покрытия погружного нефтепромыслового оборудования для защиты от коррозии, асфальтосмолопарафиновых и солевых отложений: Обзор. *Известия вузов. Порошковая металлургия и функциональные покрытия*. 2025;19(1):58–74. <https://doi.org/10.17073/1997-308X-2025-1-58-74>

Introduction

Oil and gas fields are gradually entering the late stage of development, which is accompanied by increasing complexity in operational processes [1; 2]. At this stage, the risks of equipment failures rise significantly due to a number of factors specific to this phase of the field's lifecycle. Various complicating factors act jointly and interdependently at oilfields. However, each well typically has one dominant complication type, which serves as the primary cause of most failures. The distribution of so-called complicated wells across different production companies by their main complication type provides valuable insights into the scale and actual prevalence of corrosion-related issues within the industry.

The structure of complicated wells at LLC RN-Pur-neftegaz includes only 18 % of wells prone to the formation of asphalt-resin-paraffin deposits (ARPD), 13 % of wells where failures are caused by salt deposits, and only 6 % of wells where corrosion is the primary complication. It is important to note that the number of failures related to corrosion significantly decreased only after the implementation of corrosion-resistant pipes; however, it still remains high [3]. At Udmurtneft PJSC, corrosion accounts for 39 % of well failures [4]. A substantial number (26 %) of Udmurtneft's wells are affected by ARPD, while 1 % of the wells experience complications due to inorganic salt deposits [4]. As of early 2022, the complicated well stock of PJSC Lukoil included 14,271 wells, representing 45 % of all operating artificial lift wells [5]. The structure of Lukoil's complicated well stock consists of 74.8 % of wells affected by ARPD, 9.5 % by corrosion, and 3.8 % by inorganic salt deposits [5].

Corrosion is a fundamental issue in any industry dealing with chemically active environments. In oil and gas production, its consequences include the irreversible loss of pipe metal [6], costs associated with equipment replacement, lost profits due to well down-

time during repairs, as well as expenses related to mitigating accident consequences and maintenance of corrosion protection systems. All these factors increase production costs and reduce field profitability.

Globally, corrosion results in substantial direct and indirect financial losses [7–9]. In Russia, annual metal losses due to corrosion amount to 12 % of the country's total metal reserves, which is equivalent to the same share of the annual steel production. Approximately 10 million tons out of the 70 million tons produced annually are lost to corrosion, which translates into financial losses of USD 8 billion. Nationwide, 400,000–500,000 tons of steel are used annually to replace various types of pipelines [6].

Complications associated with the formation of ARPD [10–13] and inorganic salt deposits are no less significant, as they lead to partial or complete blockages of the internal pipe cross-section, resulting in reduced production rates or even cessation of oil extraction [14–16].

The most effective solution to combat these types of complications today is the application of various functional coatings – polymeric, ceramic, and metallization – depending on the type of equipment being protected [17].

This article reviews the application of various functional coatings for oil and gas equipment aimed at counteracting complicating factors.

Functional internal coatings for tubing

Tubing (TBG) is one of the key components of submersible oilfield equipment, susceptible to corrosion, asphalt-resin-paraffin deposits (ARPD), and salt deposits on its inner surface, which is in direct contact with the extracted medium. The housings of submersible electric motors (SEM) and electric submersible pumps (ESP) are exposed to the extracted media from their external surface. It should be noted that cases of contact

between the medium and the external surface of tubing are possible; however, on the one hand, such occurrences are rare, and on the other hand, the only available method of protection against corrosion in such cases is the use of alloyed corrosion-resistant steels.

Currently, there is no established classification system for internal tubing corrosion protection methods. Therefore, the author has proposed a classification (Fig. 1) that summarizes various approaches – ranging from traditional steel alloying to the development of duplex coatings [18; 19], which combine sacrificial and barrier properties.

Austenitic stainless steel tubing is not manufactured due to its extremely high cost combined with relatively low strength properties [20]. However, this class of steel has found its application in the production of liners (thin-walled internal metal tubes) used in bimetallic pipes [21]. The installation of liners in production and tubing pipes can be performed using several methods; however, currently, only two methods have been implemented in Russia [22–24]. In both cases, the first step involves inserting a liner with a diameter smaller than the internal diameter of the tubing. The liner is secured by mechanical interference through roller expansion [25] or by a hydraulic method [26]. In both methods, the gap between the liner and the tubing is eliminated, and the resulting interference fit ensures a stable liner position under all permissible operational loads. In large-diameter pipes (used for trunk oil and gas pipelines), a metallurgical method is applied, forming a diffusion transition zone between the metals.

Various austenitic stainless steels with different chemical compositions can be used for this technology.

The most commonly used steels are AISI 304, being the most cost-effective in this class; however, their corrosion resistance is often insufficient for highly mineralized environments. In such cases, more highly alloyed steels such as AISI 316, 316L, or 825 are required.

The advantages of this technology include the relatively low cost of tubing with liners compared to pipes made entirely from liner material; the ability to achieve near-absolute corrosion resistance by selecting an appropriate liner material; and the absence of temperature limitations typical for polymer coatings. However, the disadvantages of this technology include higher costs compared to low-alloy steel pipes, low efficiency if the liner material is improperly selected, lack of reliable methods for protecting the annular space of couplings at pipe ends, and an increase in tubing string weight by 8–11 %.

It is also important to highlight the risk of pitting corrosion in austenitic steels when exposed to environments with high chloride ion (Cl^-) content. Fig. 2 shows an example of the operation of a tubular component made from AISI 316 steel as part of a pipeline transporting seawater. The corrosion rate of the component was 12.8 mm/year. This phenomenon is most likely to occur due to β -phase precipitation, such as during cold plastic deformation. Before operating in such environments, it is mandatory to conduct pitting corrosion resistance testing in ferric chloride solution according to GOST 9.912–89.

Bimetallic pipes with an internal liner can be considered a specialized type of coated pipe, with the liner serving as a protective barrier. The mechanism by which the liner provides protection is similar to that

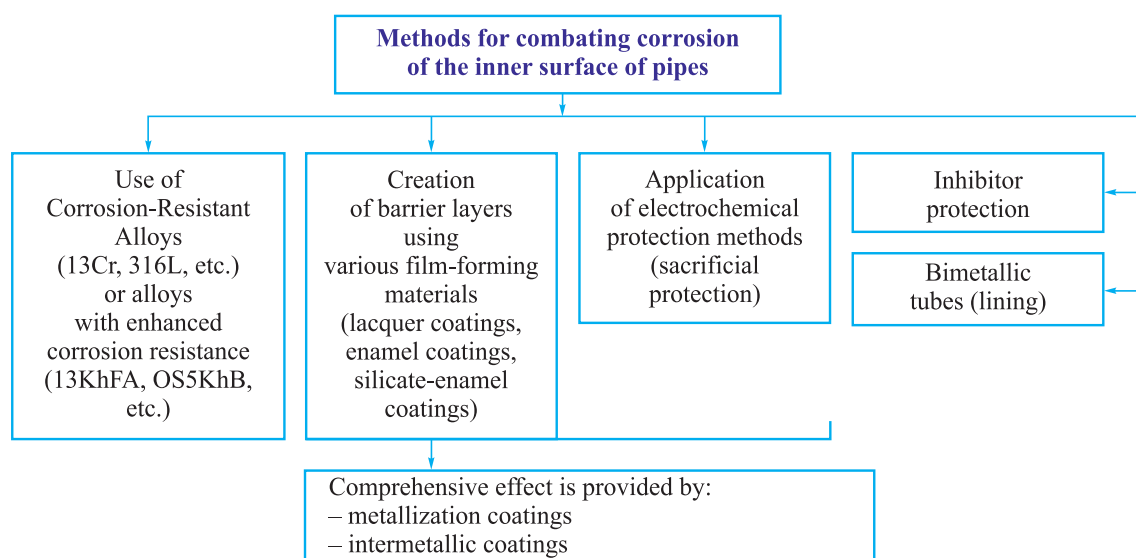


Fig. 1. Classification of methods for combating corrosion on the inner surface of tubing strings

Рис. 1. Классификация методов борьбы с коррозией внутренней поверхности насосно-компрессорных труб

of conventional coatings, resulting in comparable performance requirements for both. Based on an analysis of liner steel corrosion, manufacturing technologies, and operational factors, GOST 70926–2023, “Tubing with Internal Liner. Technical Specifications”, was developed under the author’s supervision. Pipes produced in accordance with this standard have been in trouble-free operation across Russia for more than three years. Their use commenced even before the official publication of GOST 70926–2023, following the completion of research and development activities that formed the basis for the standard. Condition monitoring has shown no signs of corrosion damage to the liner. The service life of these pipes is primarily constrained by factors such as length reduction due to repeated thread cutting, external surface corrosion, and mechanical damage.

The most commonly used corrosion-resistant steels are alloyed with chromium in concentrations of 13 % or more. According to Schaeffler’s rule, their protection mechanism is based on the formation of a passive chromium oxide film on the surface, which is resistant to interaction with corrosive gases dissolved in the produced fluid. Traditionally, Cr13-grade steel pipes are considered the benchmark for corrosion resistance; however, their widespread adoption is limited by their high cost. Certain factors significantly reduce the service life of such pipes and lead to premature failure, all of which are associated with the degradation of the passive layer. Since the produced fluid lacks

oxygen, repassivation is not possible. The operational limitations of Cr13-grade pipes include the following: acid treatments, mechanical wear (including erosion), and cable insulation failure. In the presence of hydrogen sulfide in the well and acid treatments, sulfide stress corrosion cracking (SSCC) may occur [27; 28], which is typical for low-alloy steels with a hardness exceeding 22 HRC.

The term “steel with enhanced corrosion resistance” was introduced in Russia in the mid-2000s. This class includes steels alloyed with 0.5–1.0 % Cr, often with the addition of niobium or vanadium to refine the grain structure, as well as copper. However, according to GOST 5272–50, these steels are classified as low-alloy and are not considered corrosion-resistant (it is worth noting that the later and currently relevant edition, GOST 5272–68, does not include a corrosion resistance scale, nor is the term “steel with enhanced corrosion resistance” mentioned in other regulatory documents). The experience of their implementation has been mixed, and given their relatively high cost, they are not considered an optimal solution [29].

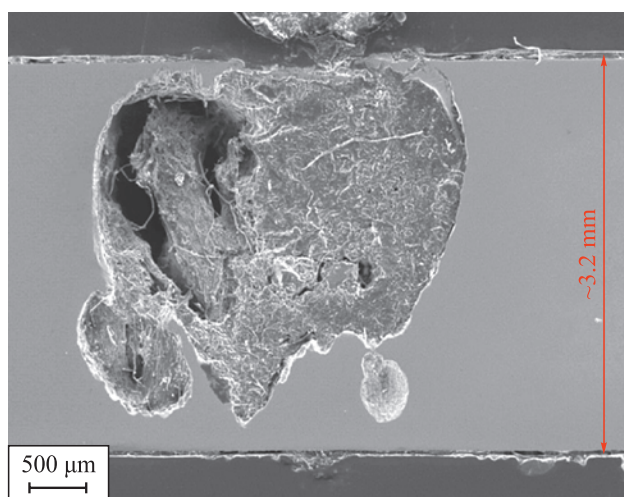
The optimal solution for protecting the inner surface of tubing from complicating factors is the use of various functional coatings [17; 30; 31]. Despite the wide variety of available coatings, in practice, only silicate-enamel (SEC), polymer, and duplex (intermetallic layer + polymer layer) coatings are commonly applied.

Silicate-enamel coatings are formed from a slurry prepared using frits of MK-5 and MK-5U grades with the following composition (wt. %): 0.5–3.5 Al_2O_3 ; 10.0–16.0 B_2O_3 ; 8.0–16.0 Na_2O ; 0.5–5.0 K_2O ; 2.0–5.0 Li_2O ; 2.0–8.0 CaO ; 0.1–1.0 MgO ; 3.0–6.0 TiO_2 ; 0.5–5.0 MnO_2 ; 0.3–2.0 NiO ; 0.2–2.0 CuO ; 0.3–1.5 CoO ; 0.1–1.5 Fe_2O_3 ; and 0.5–4.0 F (in excess of 100 %). The coating can be applied using either a liquid or powder method, followed by firing at temperatures ranging from 850 to 950 °C. During the firing process, gases are released, which, upon cooling, result in the formation of porosity, often characterized by through-pores.

Despite their excellent resistance to ARPD formation, the use of tubing with silicate-enamel coatings (SEC) in the oil and gas industry has been declining, with the total volume not exceeding several thousand tons as of 2024. This reduction is attributed to several significant limitations:

- through-pore porosity in single-layer coatings (Fig. 3), which leads to severe pitting corrosion beneath the pores;

- high brittleness, which imposes restrictions on mechanical impacts and thread tightening torque; exceeding these limits can result in enamel chipping, particularly in the nipple area at the pipe ends;



C	Si	Mn	Cr	Ni	S	P	Cu	Mo
0.03	0.51	1.77	16.8	9.19	0.05	0.01	0.43	1.80

Fig. 2. Longitudinal section of the tubular component wall made of AISI 316 steel after 3 months of operation as part of a seawater pipeline

Рис. 2. Продольное сечение стенки трубчатого изделия из стали AISI 316 после 3 месяцев эксплуатации в составе трубопровода морской воды

- heat curing requirements at 850–950 °C, which prevent the use of heat-treated steel pipes, as such temperatures degrade their mechanical properties;

- higher costs compared to polymer coatings, primarily due to the need for high-temperature processing.

Advancements in polymer coatings, particularly the development of multifunctional coatings that combine anti-corrosion properties with resistance to ARPD formation, have largely negated the advantages of tubing with silicate-enamel coatings.

At present, the primary method for protecting the inner surface of tubing from complicating factors is the application of polymer coatings [32–35]. Their widespread adoption began in the early 2000s with the emergence of manufacturers such as MajorPack and Hilong. Although several production lines existed in Russia before this period, their products were not widely used. In the early stages of development, various film-forming coatings were applied, such as the polyurethane-based PolyPlex coating. However, it demonstrated an extremely short service life (typically less than 30 days), leading to its swift abandonment within the industry.

Polymer coatings offer several advantages over other protective methods against complicating factors. They provide extended service life for equipment and pipelines by protecting against corrosion and wear, significantly prolonging operational lifespan and reducing repair and replacement costs. Maintenance costs are also lower, as polymer coatings minimize the need for frequent servicing and repairs. Additionally, they enhance oil production and transportation efficiency by reducing friction, preventing deposits such as ARPD and inorganic salts, and improving thermal insulation,

which leads to greater system performance. The risk of failures is reduced, as effective corrosion and wear protection helps prevent breakdowns associated with equipment deterioration. Furthermore, polymer coatings offer significant cost savings by extending service life and decreasing maintenance and repair expenses. Lastly, they provide environmental benefits by reducing emissions and leaks through improved corrosion resistance and enhanced equipment protection [36; 37].

The classification of coatings used to protect the inner surface of tubing is shown in Fig. 4.

There are several methods for applying internal polymer coatings, with the most commonly used being airless spraying and electrostatic application. The airless spraying method applies liquid coatings to the surface without the use of an air stream. Instead, high pressure forces the material through a nozzle, breaking it into fine droplets that uniformly coat the surface. This technique provides higher efficiency and better application quality compared to conventional air spraying and allows for the application of high-viscosity coatings with 100 % solids content. Electrostatic application, on the other hand, involves applying a layer of polymer powder to the product's surface using an electrostatic field. In this process, the polymer particles are electrically charged and attracted to the surface, which has an opposite charge, ensuring a uniform and durable coating [38].

Initially, the effectiveness of internal coatings was questioned, as early applications utilized cold-cured epoxy resins based on Bisphenol A and epoxy novolac resin in an approximate 2:1 ratio (Fig. 5). The fillers and pigments commonly used in these coatings included micronized barite (BaSO_4), aerosil (fumed silica, SiO_2), titanium dioxide (TiO_2), and talc ($\text{Mg}_3\text{Si}_4\text{O}_{10}(\text{OH})_2$). Their concentrations varied depending on the formulation, but the total filler content typically did not exceed 50 %. These coatings exhibited relatively low glass transition temperatures (usually below 60 °C), limited chemical resistance, and restricted operating temperature ranges. They were applied using dual-component airless spray systems.

The degradation of pipes with the aforementioned type of coatings (Fig. 6), as evidenced by more than 40 expert studies conducted under the author's supervision, highlights the low barrier properties of these coatings. The service life before the onset of blistering and delamination typically does not exceed 2–3 years for field pipelines. The use of such coatings for tubing protection is viable only at relatively low temperatures (up to 40 °C) and in cases where complications are limited to ARPD. Even under these conditions, their service life rarely exceeds 1 year.

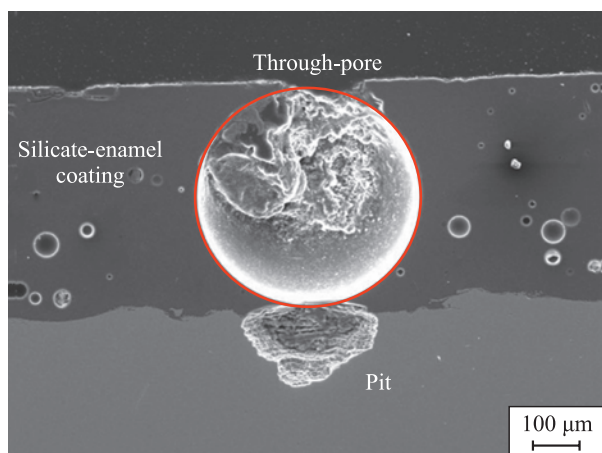


Fig. 3. Formation of pitting corrosion through the mechanism of carbon dioxide corrosion in a silicate-enamel coating

Рис. 3. Образование язвы по механизму уголекислотной коррозии в силикатно-эмалевом покрытии

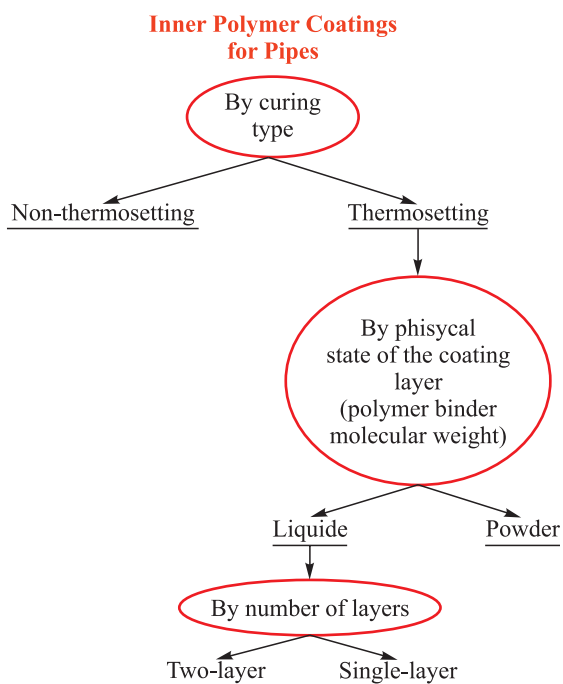


Fig. 4. Classification of polymer coatings used to protect the inner surface of tubing

Рис. 4. Классификация полимерных покрытий, применяющихся для защиты внутренней поверхности НКТ

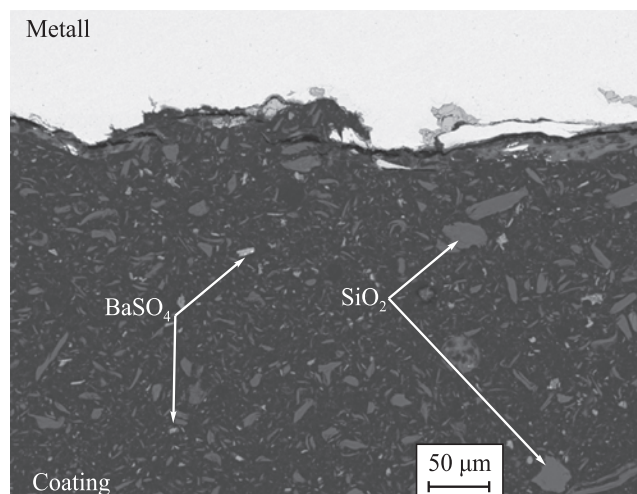


Fig. 5. Structure of a typical liquid non-thermosetting internal epoxy coating for an oil pipeline

Рис. 5. Структура типичного жидкого нетермоотверждаемого внутреннего эпоксидного покрытия нефтепроводной трубы

The liquid thermosetting epoxy novolac coating (its structure shown in Fig. 7) stands apart from other coatings due to its higher concentration of novolac resin, with an epoxy novolac resin to Bisphenol A-based resin

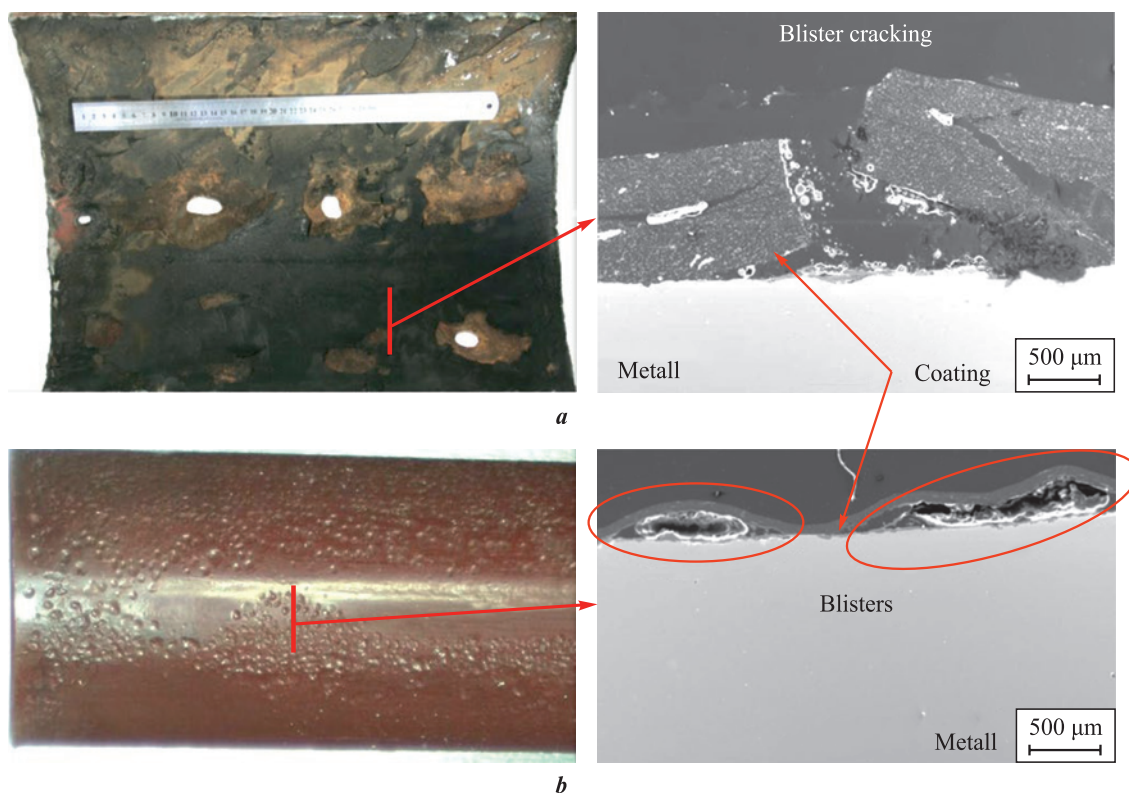


Fig. 6. Degradation of liquid non-thermosetting internal epoxy coatings used in $\varnothing 426 \times 8$ mm oil collection manifold at the Mamontovskoe field (a) and $\varnothing 73 \times 5.5$ mm production tubing at the Krapivinskoe field (b)

Рис. 6. Разрушение жидких нетермоотверждаемых внутренних эпоксидных покрытий, эксплуатировавшихся в составе нефтесборного коллектора $\varnothing 426 \times 8$ мм Мамонтовского месторождения (a) и добывающей скважины НКТ $\varnothing 73 \times 5,5$ мм Крапивинского месторождения (b)

ratio of $\geq 1:1$, and the inclusion of a reactive diluent. This diluent lowers the viscosity of the epoxy composition, facilitating easier processing while also participating in the curing reaction and becoming an integral part of the polymer matrix. The application process is similar to that of non-thermosetting liquid epoxy coatings, with the key difference being the requirement for an isothermal curing hold at 180–200 °C for at least 20 min.

This type of coating is the most commonly used for tubing protection, as it delivers the required performance characteristics at relatively small thicknesses ($\sim 150 \mu\text{m}$). It also enables the application of two-layer coatings, allowing for formulation optimization at smaller production facilities. The industry-wide use of these coatings is estimated to be 85–90 %.

In the past three years, two-layer powder coatings have been increasingly adopted, consisting of a primer layer with a thickness of 5–40 μm and a top layer of $\geq 350 \mu\text{m}$ made from a powder composite material (Fig. 8). The primer is a paint-and-lacquer material composed of a blend of high-molecular-weight epoxy and phenol-formaldehyde resins, butyl cellosolve, toluene, and, in most cases, iron oxide pigment, which actively reacts with hydrogen sulfide during operation or testing. The powder coating is applied electrostatically to a pre-heated surface at a temperature of 160–200 °C. Its composition includes high-molecular-weight epoxy resin and a significant amount of fillers (up to 70 %), with a composition similar to those described earlier. This technology offers several advantages, including superior barrier properties, lower costs compared to other pro-

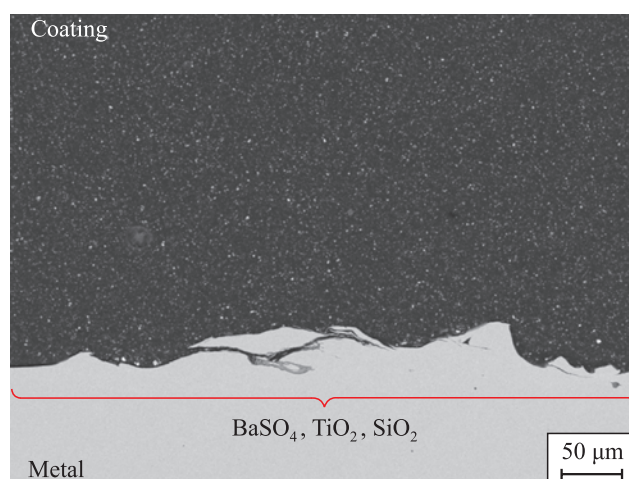


Fig. 7. Structure of a typical liquid thermosetting internal epoxy coating for an oil pipeline

Рис. 7. Структура типичного жидкого термоотверждаемого внутреннего эпоксидного покрытия нефтепроводной трубы

tective methods, high process efficiency and maintainability, and the ability to apply multifunctional coatings.

Additionally, the absence of solvent evaporation during application makes this method more environmentally friendly and safer for production workers, which is why it is the only permitted option in most countries. The production volume of tubing with internal functional coatings is expected to exceed 100,000 tons by the end of 2024 (based on the weight of $\varnothing 73 \times 5.5$ mm tubing) and continues to grow at a rate of 8–12 % per year.

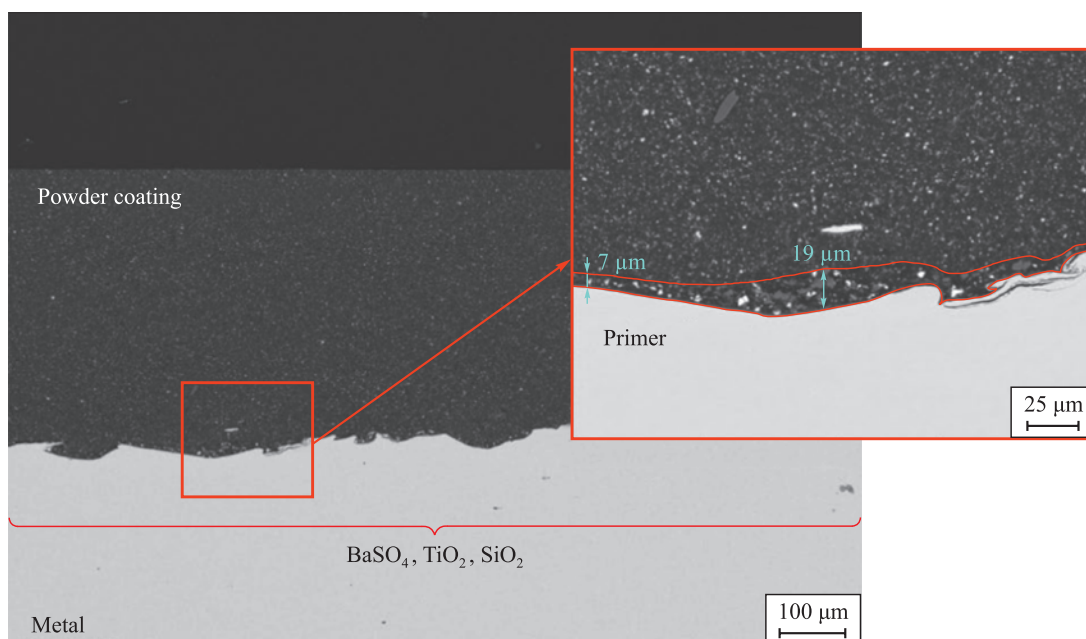


Fig. 8. Structure of a typical two-layer powder coating with an epoxy-phenolic primer

Рис. 8. Структура типичного двухслойного порошкового покрытия с эпоксифенольным праймером

A relatively new technological solution is the use of duplex coatings, where the first layer consists of thermodiffusion zinc (TDZ), forming Fe–Zn intermetallic compounds, followed by one or more polymer layers [39]. These coatings demonstrate higher corrosion resistance in hydrogen sulfide-containing environments, as confirmed by laboratory autoclave exposure tests conducted over 2352 h, compared to the standard test duration of 240 h specified by GOST 58346–2019. The test involved exposing samples with two types of coatings in an autoclave under a partial gas pressure of H_2S – 1 MPa, N_2 – 9 MPa at a temperature of 80 °C. The results showed the presence of iron sulfide at the metal-coating interface in pipes without the TDZ layer, whereas no corrosion products were detected on duplex-coated pipes (Fig. 9). It should be noted that the autoclave testing methods used in this experiment and outlined in GOST 58346–2019 are based on previous research [40; 41]. The widespread adoption of autoclave testing methods has led to a significant increase in the service life of tubing with internal coatings. Although exact statistical data is not publicly available, sources accessible to the author indicate that the average service life has increased from 418 to 786 days. However, the disadvantages of duplex coatings include higher costs compared to conventional polymer-coated pipes and the limited availability of reliable solutions for protecting the external surface from corrosion.

A key trend in the development of tubing with internal coatings is the creation of multifunctional coatings that combine anti-corrosion and anti-abrasion properties while also preventing ARPD and inorganic salt deposition [42–45]. Until recently, advancements in coating formulations and application technologies – such as the introduction of non-thermal microwave

treatment [46] – were limited by the lack of standardized laboratory testing methods. The applicability of each coating was primarily determined through field pilot tests (FPT), which typically take about one year to complete.

To address the challenge of simulating ARPD formation under controlled conditions, two circulating test benches were developed and manufactured under the author's supervision [47; 48]. The test medium used in these test benches is an oil emulsion sampled from wells affected by ARPD, further enriched with deposits obtained during cleaning operations. The design of the test benches allows for adjustments in the composition of the test medium, flow rate, medium temperature, and the external surface temperature of the sample. The temperature difference across the inner surface of tubing samples – used as test specimens – facilitates the formation of deposits. The capabilities of these benches cover a wide range of well conditions, from low- to high-production wells, with varying temperature regimes influencing ARPD formation.

Research findings indicate that parameters such as surface roughness, paraffin adhesion to a dry surface, and the contact angle of distilled water on a dry surface do not provide a reliable assessment of a surface's resistance to ARPD deposition. The laboratory method for determining the contact angle by measuring the spreading of an oil droplet in water on the coating surface has shown the highest correlation with test bench results [49]. The best ARPD resistance results were demonstrated by hydrophilic surfaces; however, they exhibited poor corrosion resistance. Therefore, the use of multifunctional coatings or two-layer systems is required to achieve balanced performance.

Experimental studies have established a correlation between ARPD deposition and flow rate (Reynolds number), allowing for the ranking of coatings based on their resistance to ARPD. Silicate-enamel coatings provide the highest resistance, followed by polymer coatings, while bare steel samples show the lowest performance (Fig. 10). The obtained results align with data from field pilot tests and the operational performance of these coating types across various oilfields.

Another important step in expanding the application of functional polymer coatings was the development of a test bench (Patent No. RU2825169C1) and a methodology to evaluate the effectiveness of coatings in preventing inorganic scale formation [50]. The goal of the test trials was to identify the coating with the highest resistance to scaling under oilfield conditions. The coatings' resistance to scaling was assessed based on the mass of inorganic scale formed

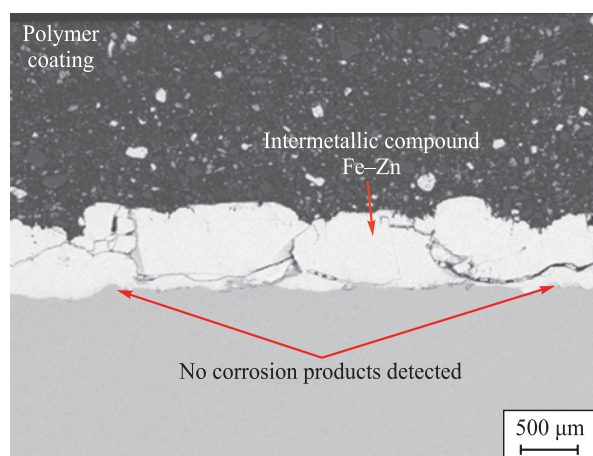


Fig. 9. Surface condition of the metal–intermetallic layer (Fe–Zn) after an autoclave test for 2352 h

Рис. 9. Состояние поверхности «металл–интерметаллидный слой (Fe–Zn)» после автоклавного теста в течение 2352 ч

on the outer surfaces of cylindrical samples and the thickness of the resulting scale layer.

The results of the test trials evaluating the resistance of coatings to gypsum-type (CaSO_4) scale formation with halite (NaCl) impurities were published in [51]. It was found that none of the tested protective coatings could completely prevent the formation of gypsum scale with halite impurities on their surface. The study [51] also concluded that the adhesion strength of the “scale–coating” interface does not play a decisive role in the anti-scaling properties of protective coatings. The findings showed that scale deposits can form even on surfaces with minimal adhesion strength. These deposits are capable of creating solid structures with little to no interaction with the surface. Further analysis in [52] compared the results of the test trials [51] with the roughness parameters of the tested protective coatings to evaluate the effect of surface roughness on scaling. A certain correlation was observed between the coating’s roughness index and the mass of the scale layer formed on it. The steel sample with the highest surface roughness exhibited the most significant increase in the scale layer mass during dynamic tests. However, the relationship between surface roughness and scale formation mass was not strictly linear [52].

Since studies [51] and [53] both found that none of the examined coatings could completely prevent scale formation, further research focused on assessing the combined use of coatings with other preventive methods. In study [54], laboratory-scale dynamic tests were conducted to evaluate the feasibility of integrating internal protective tubing coatings with scale inhibitors. The findings showed that the combined approach can provide the following benefits:

– coated surfaces had fewer crystallization centers for inorganic scale deposits compared to uncoated steel surfaces;

– during testing (with a scale inhibitor dosage of 200 g/m^3), the formed scale deposits detached more readily from the coated samples [54].

Coating of SEM and ESP housings

Various corrosion protection methods are employed to mitigate the corrosive impact and extend the service life of submersible electric motors (SEM) and electric submersible pumps (ESP). (Hereinafter, the discussion will focus solely on SEM housings; however, all conclusions are equally applicable to ESP housings.) One of the simplest and most cost-effective ways to enhance the service life of SEM housings while reducing exposure to aggressive factors under field conditions is the application of metallization coatings. Among the most widely used methods for applying such coatings are electric arc spraying (EAS) and high-velocity oxy-fuel (HVOF) spraying [55–60].

An analysis of the causes of SEM and ESP housing failures was conducted under the author’s supervision in four oil-producing regions with various complicating factors. The study revealed that corrosion of the SEM housing is the most common cause of failure [61–63]. The examination of SEM housings with metallization coatings applied using technologies implemented at pipe bases after operation allowed the identification of key causes of failure, including mechanical damage, abrasive wear of the coating, low barrier properties, and imperfections in the coating application technology (Fig. 11). Additionally, cases were identified where multiple negative factors were simultaneously present, making it impossible to determine the dominant failure mechanism. In many cases, these factors create a synergistic effect – for example, the simultaneous presence of a corrosive environment and abrasive particles results in corrosion-erosion wear, which occurs at a significantly higher rate compared to either corrosion or erosion alone [61].

The primary challenge in ensuring the operational reliability of SEM housings was the development of methodologies that simulate the effects of key complicating factors and conducting laboratory tests on various types of metallization coatings. The results, summarized in [61], confirm the feasibility of accurately modeling the destructive impact of key complicating factors through autoclave tests in H_2S - and CO_2 -containing environments [64; 65]. To determine the corrosion resistance of metallization coatings, samples must undergo autoclave exposure for 240 h. Testing typically involves the combined exposure

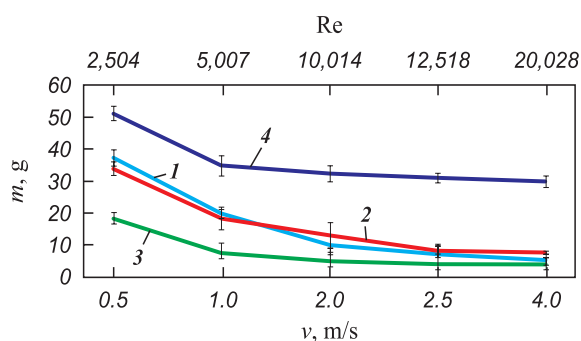


Fig. 10. Dependence of asphaltene-resin-paraffin deposit mass on the flow rate of the oil medium on various surfaces

1 – GIOTEK 110M, 2 – MPLAG17,
3 – silicate enamel MK-5, 4 – uncoated

Рис. 10. Зависимость массы выпадения АСПО от скорости потока нефтяной среды на различных поверхностях

1 – ГИОТЭК 110М, 2 – MPLAG17,
3 – силикатно-эмалевое МК-5, 4 – без покрытия

to aggressive gases (CO_2 , H_2S). In order to isolate the corrosion effects of individual system components, tests can also be conducted in environments saturated with only carbon dioxide or hydrogen sulfide. To prevent decompression effects, pressure release must take at least 10 min [64].

Electric arc spraying (EAS) was selected as the primary coating application method due to its high process efficiency and equipment mobility. EAS coatings made of stainless steel materials based on iron and nickel were applied with a target thickness of 350–500 μm . The chemical composition of the metallization coatings included various elements in specific proportions (wt. %):

– Cr ~ 6.6÷14.5; Ni ~ 4.4÷8.4; Mo ~ 2.5; Si ~ 2.0÷3.7; Fe – balance;

– Cr ~ 13.0÷16.0; Ni ~ 7.3÷9.8; Mo ~ 3.5; Si ~ 2.9÷3.7; Al ~ 1.0; Fe – balance;

– Cr ~ 17.7÷18.6; Ni ~ 8.5÷8.9; Ti ~ 0.6; Si ~ 0.5÷0.7; Fe – balance.

The physical and mechanical properties of EAS coatings were found to be low due to significant porosity and oxide layers between particles. Metallization

coatings applied using iron-based wires (Fig. 12, *a*) without additional impregnation or an external polymer layer proved to be ineffective against corrosive environments. In contrast, nickel-based wire coatings (Fig. 12, *b*) demonstrated resistance in acidic environments but lacked sufficient protection against CO_2 - and H_2S -saturated conditions, indicating their inability to provide long-term protection. To reduce particle oxidation during spraying, an argon protective atmosphere was used (Fig. 12, *c*, *d*), which resulted in a 55 % improvement in physical and mechanical properties compared to coatings applied in open air; however, corrosion resistance remained unsatisfactory [66].

Impregnation materials are commonly used to seal coating pores [67; 68]. The study utilized polymer-based impregnating materials, including epoxy-phenolic, acrylic, and polytetrafluoroethylene compositions with thicknesses of 70–150 μm . The application of impregnation materials improved the corrosion resistance of the metallization layer; however, even minor damage to the impregnated coating on iron-based layers resulted in rapid degradation. Therefore, the use of iron-based coatings with impregnation was deemed impractical.

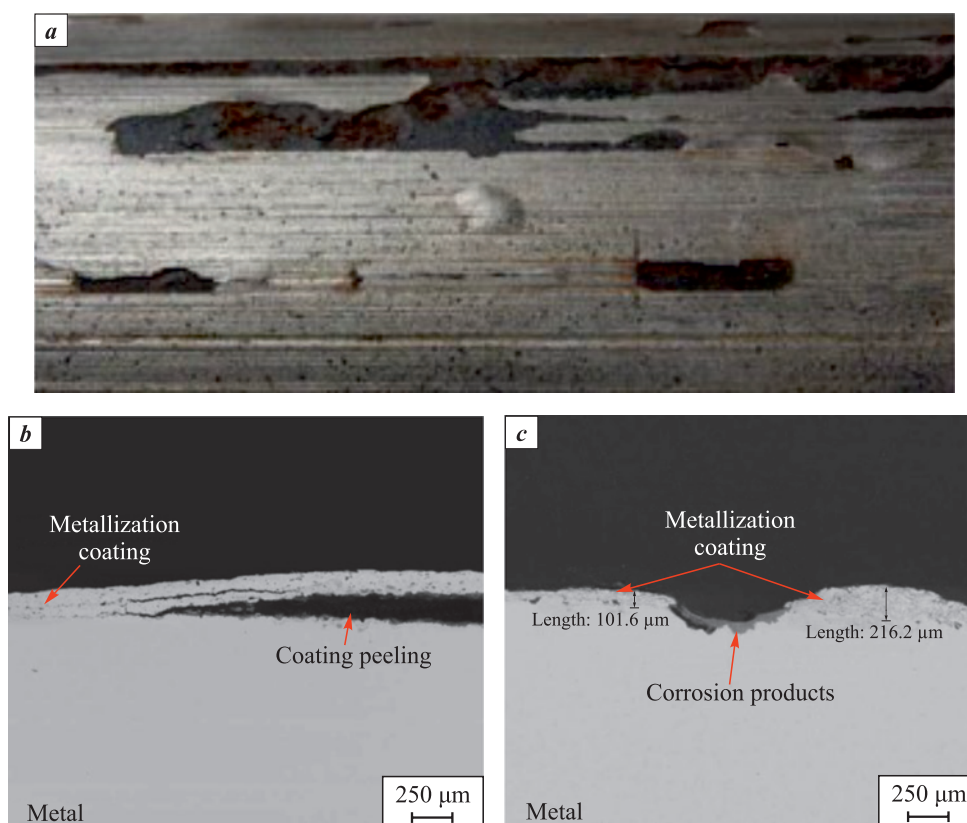


Fig. 11. External view of the SEM housing with peeling and blistering of the metallization coating (*a*); destruction in the form of swelling of the coating (*b*); formation of corrosion products at the site of local damage to the coating (*c*)

Рис. 11. Внешний вид корпуса ПЭД с отслоениями и вздутиями металлизационного покрытия (*a*); разрушение в виде вздутия покрытия (*b*); образование продуктов коррозии в месте локального повреждения покрытия (*c*)

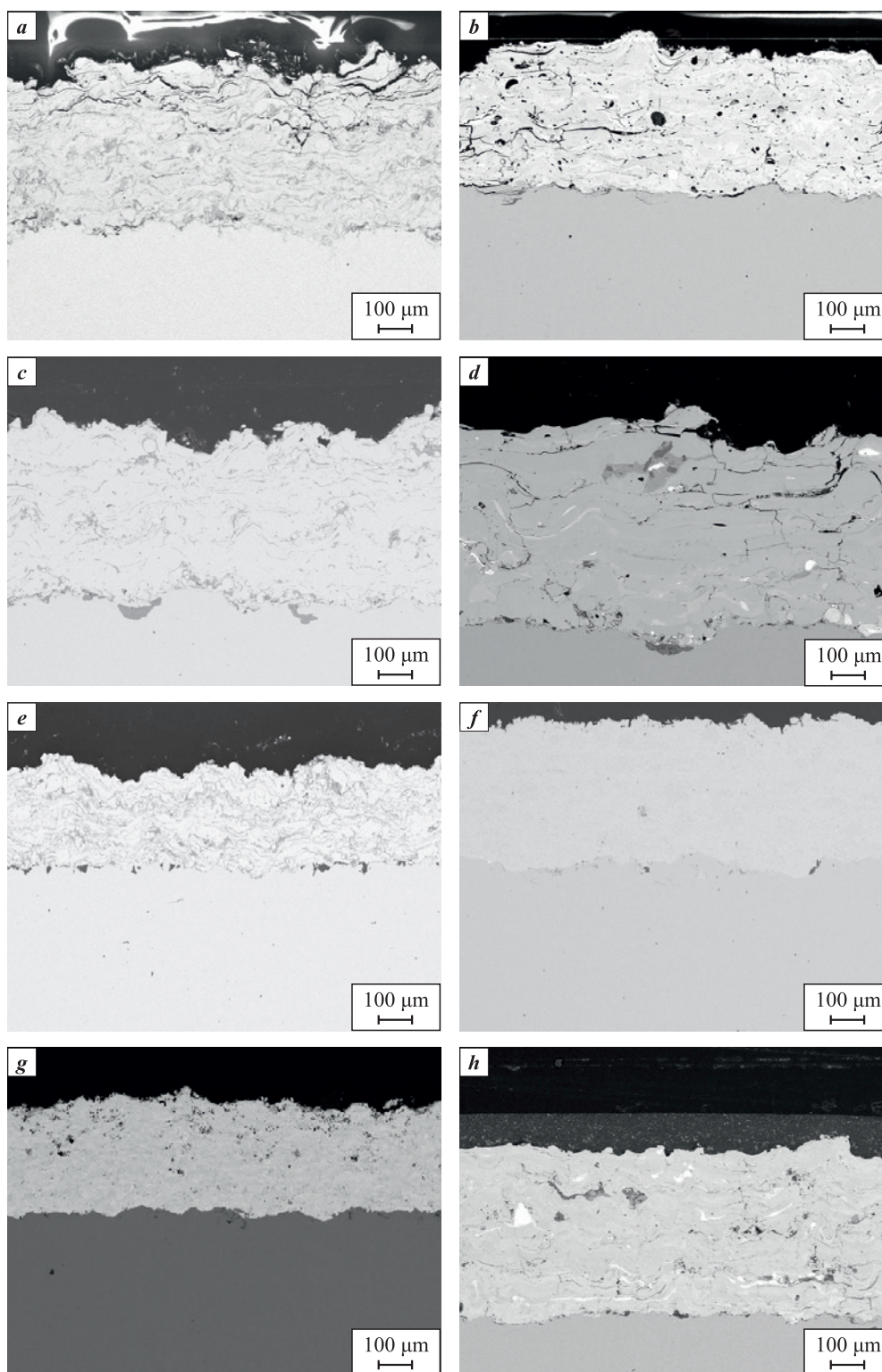


Fig. 12. Microstructure of the coating

a – EDM, iron-based wire; *b* – EDM, nickel-based wire; *c* – EDM in an argon environment, iron-based wire;
d – EDM in an argon environment, nickel-based wire; *e* – АДМ, iron-based wire; *f* – VSGPN, self-fluxing nickel-based powder;
g – VSGPN, tungsten carbide powder in a cobalt matrix; *h* – EDM, nickel-based wire impregnated with epoxy-novolac resin

Рис. 12. Микроструктуры различных покрытий

a – ЭДМ, проволока на основе железа; *b* – ЭДМ, проволока на основе никеля;
c – ЭДМ в среде аргона, проволока на основе железа; *d* – ЭДМ в среде аргона, проволока на основе никеля;
e – АДМ, проволока на основе железа; *f* – ВСГПН, самофлюсующийся порошок на основе никеля;
g – ВСГПН, порошок на основе карбида вольфрама в кобальтовой матрице;
h – ЭДМ, проволока на основе никеля с пропиткой на основе эпоксинаволачной смолы

To increase the spraying flame temperature and achieve greater particle melting, activated arc spraying (AAS) was employed in a propane-air environment [69]. The use of AAS with iron-based wires improved the physical and mechanical properties of the coating (Fig. 12, e). However, due to the combustion of the propane-air mixture during application, the metallization layer still contained oxide layers that were not resistant to aggressive environments, making AAS coatings unsuitable in their pure form. The application of an epoxy novolac resin-based impregnation helped to limit the access of corrosive media to the metallization layer, thus preventing its destruction. However, in areas with artificially introduced defects such as “scratches”, the metallization layer deteriorated, and corrosion products were observed on the substrate.

Using high-velocity oxy-fuel (HVOF) spraying, self-fluxing powder materials based on nickel (Fig. 12, f) and tungsten carbide in a cobalt matrix (Fig. 12, g) were applied with a thickness of 300–350 µm. These coatings demonstrated high physical and mechanical properties and sufficient resistance to corrosive environments. The HVOF-applied layer exhibited higher density with fewer oxide films between particles, which was achieved due to the higher particle velocity and shorter exposure time to the gas-oxidizing environment compared to EAS. The use of tungsten carbide-based powder materials provided coatings with excellent hardness and wear resistance; however, high porosity allowed aggressive media to penetrate and lead to substrate degradation.

Based on the results of the conducted studies, the following conditions were found to provide satisfactory outcomes:

- a nickel-based metallization layer containing ~18 % chromium and ~13 % molybdenum, applied using EAS, followed by impregnation with an epoxy novolac resin composition with a thickness of 70–150 µm;

- a nickel-based self-fluxing powder metallization layer containing ~16 % chromium, applied using HVOF spraying.

It is noteworthy that with comparable properties, the cost of the HVOF-applied coating is lower than that of EAS.

Conclusion

The introduction of anti-corrosion coatings for protecting submersible equipment used in oil production, particularly the inner surface of tubing, began in the late 1990s with the adoption of polymer coat-

ings. The extension of service life was accompanied by advancements in coating formulations, transitioning from Bisphenol A-based epoxy resins to epoxy novolac resins, as well as improvements in application methods. These developments led to the introduction of thermal curing at temperatures of 170–200 °C. A significant milestone in this evolution was the development of two-layer systems that combined high barrier properties with resistance to ARPD and inorganic salt deposition. Currently, manufacturers of protective coatings face the challenge of developing multifunctional single-layer coatings that combine these protective capabilities.

At the same time, continuous improvements have been made to the composition and structure of metallization coatings. Traditional coatings applied using EAS with iron-based wires – characterized by overall porosity levels of up to 10 % and the presence of oxide films along particle boundaries – failed to provide adequate barrier protection. This resulted in the formation of corrosion products at the metal-coating interface, leading to rapid deterioration. Two parallel approaches have been pursued to enhance the operational reliability of such coatings: developing high-temperature polymer impregnations to seal porosity in the upper layers, and adopting HVOF spraying technology, which enables the production of non-porous, high-alloy corrosion-resistant coatings. While the scientific basis for these methods has been established, their full-scale implementation at production facilities is still required, along with efforts to replace imported components with domestically produced alternatives.

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